

170-14

CRYSTALLIZATION OF SOME SILICATE GLASSES

Thesis Submitted

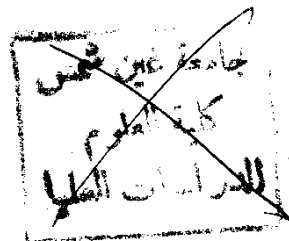
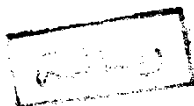
To

The Faculty of Science, Ain-Shams University

By

AHMED RASHAD BADAWY EL-GHANNAM

(B.Sc.)



In Partial Fulfillment of The Requirements For The Degree

of

Master of Science

In Chemistry



A Contribution From The National Research Centre

Dokki, Cairo, EGYPT

1987

Handwritten signatures and stamps.

548.5

A. R

ACKNOWLEDGEMENTS

*The author is grateful to Prof. Dr. **A.M. AZZAM**, Chem. Dept. Ain-Shams Uni. for his kind interest and fruitful advice.*

*The author wishes to express his deep gratitude and appreciation to Prof. Dr. **A.A. OMAR**, Head of Glass, Refractories and Building Materials Lab. National Research Centre, for suggesting the subject, supervision and sincere efforts in persuing the research prescribed.*

*The author also expresses his great indebtedness to Dr. **A.W.A. EL-SHENNAWI**, Ass. Prof. Glass, Refractories and Building Materials Lab. National Research Centre, for his supervision, valuable assistance, sincere continuous support and fruitful discussions.*

*Thanks are also due to Prof. Dr. **A.F. ABBAS**, N.R.C., for his kind interest and valuable advice especially during the first stages of this work.*

-0000000-

*



CONTENTS

	Page
ABSTRACT.....	i-ii

PART ONE

INTRODUCTION AND EXPERIMENTAL

CHAPTER I : SCIENTIFIC BACKGROUND AND LITERATURE REVIEW.

1.1. INTRODUCTION.....	1
1.2. SCIENTIFIC BACKGROUND.....	5
1.2.1. Crystallization of Glass.....	5
i) Nucleation.....	5
ii) Crystal growth.....	12
1.2.2. Crystallization Controllers.....	13
i) Base composition.....	13
ii) Nucleating catalysts.....	15
iii) Heat treatment.....	16
1.3. LITERATURE REVIEW OF CRYSTALLINE PHASES RELATED TO THE PRESENT WORK.....	18
1.3.1. Spodumene ($\text{LiAlSi}_2\text{O}_6$) and Related Phases.....	20
1.3.2. Lithium Zinc Orthosilicate ($\text{Li}_2\text{ZnSiO}_4$) and Related Phases.....	26
1.3.3. Lithium Magnesium Orthosilicate ($\text{Li}_2\text{MgSiO}_4$) and Related Phases.....	32
1.4. AIM OF THE WORK.....	35

CHAPTER II : EXPERIMENTAL TECHNIQUE.

2.1. GLASS PREPARATION.....	36
2.2. HEAT TREATMENT SCHEDULES.....	37
2.2.1. Single-Stage Heat Treatment.....	37
2.2.2. Double-Stage Heat Treatment.....	39
2.3. MICROSCOPY.....	39
2.3.1. Polarizing Microscopy.....	39
2.3.2. Electron Microscopy.....	40
2.4. X-RAY DIFFRACTION ANALYSIS.....	40
2.5. DIFFERENTIAL THERMAL ANALYSIS (DTA).....	41
2.6. DILATOMETRY.....	46

PART TWO
RESULTS AND DISCUSSION

CHAPTER III : CRYSTALLIZATION OF $\text{Li}_2\text{ZnSiO}_4$ - $\text{LiAlSi}_2\text{O}_6$
GLASSES.

3.1. . CRYSTALLIZATION OF BASE GLASSES.....	50
3.1.1. DTA of Base Glasses.....	52
3.1.2 Thermal Treatments.....	54
i) Heat treatment of G2Z.....	56
ii) Heat treatment of G4Z.....	60
iii) Heat treatment of G5Z.....	63
3.1.3. Discussion.....	68
3.2. CRYSTALLIZATION OF GLASSES CONTAINING TiO_2 AS NUCLEANT.....	79
3.2.1. DTA of Glasses Containing TiO_2	79
3.2.2. Thermal Treatments.....	80
3.2.3. Discussion.....	89
3.3. CRYSTALLIZATION OF GLASSES CONTAINING ZrO_2 AS NUCLEANT.....	96
3.3.1. DTA of Glasses Containing ZrO_2	96
3.3.2. Thermal Treatments.....	98
3.3.3. Discussion.....	104

CHAPTER IV : CRYSTALLIZATION OF $\text{Li}_2\text{MgSiO}_4$ - $\text{LiAlSi}_2\text{O}_6$
GLASSES.

4.1. CRYSTALLIZATION OF BASE GLASSES.....	110
4.1.1. DTA of Base Glasses.....	111
4.1.2. Thermal Treatments.....	113
i) Heat treatment of G2M.....	113
ii) Heat treatment of G4M.....	117
iii) Heat treatment of G6M.....	119
4.1.3. Discussion.....	122
4.2. CRYSTALLIZATION OF GLASSES CONTAINING TiO_2 AS NUCLEANT.....	130
4.2.1. DTA of Glasses Containing TiO_2	130

4.2.2. Thermal Treatments.....	132
4.2.3. Discussion.....	138
4.3. CRYSTALLIZATION OF GLASSES CONTAINING ZrO_2 AS NUCLEANT.....	143
4.3.1. DTA of Glasses Containing ZrO_2	143
4.3.2. Thermal Treatments.....	144
4.3.3. Discussion.....	148
CHAPTER V : THERMAL EXPANSIVITIES OF THE GLASSES AND THEIR CORRESPONDING GLASS-CERAMICS.	
5.1. THERMAL EXPANSIVITIES OF Li_2ZnSiO_4 - $LiAlSi_2O_6$ GLASSES AND THEIR CORRESPONDING GLASSES-CERAMICS.....	150
5.1.1. Effect of Nucleators.....	155
5.2. THERMAL EXPANSIVITIES OF Li_2MgSiO_4 - $LiAlSi_2O_6$ GLASSES AND THEIR CORRESPONDING GLASS-CERAMICS.....	159
5.2.1. Effect of Nucleators.....	162
CHAPTER VI : SUMMARY AND CONCLUSIONS.	165
REFERENCES.....	171
SUMMARY IN ARABIC.....	

-ooo0ooo-

*

ABSTRACT

The effect of compositional variation, thermal treatment and nucleation catalysts TiO_2 and ZrO_2 on the nature, type and stability of the crystallizing phases as well as the resulting microstructures is described for some stoichiometric glass compositions within the systems $\text{Li}_2\text{ZnSiO}_4$ - $\text{LiAlSi}_2\text{O}_6$ and $\text{Li}_2\text{MgSiO}_4$ - $\text{LiAlSi}_2\text{O}_6$ to obtain glass-ceramic materials. Differential thermal analysis, X-ray diffraction, dilatometry, polarizing and electron-microscopy were used for studying the crystallization process and characterization of the resulting materials.

Fine-grained glass-ceramic materials with favourable thermal properties were resulted from glasses rich in $\text{Li}_2\text{ZnSiO}_4$ or $\text{Li}_2\text{MgSiO}_4$ components. Transparent glass ceramics with ultrafine microstructures could also be obtained from glasses within the $\text{Li}_2\text{ZnSiO}_4$ - $\text{LiAlSi}_2\text{O}_6$ system .

The crystallization of Zn-containing glasses begins with the formation of proto β -and/or β'_{II} - $\text{Li}_2\text{ZnSiO}_4$ followed by, or concomitant with, β -eucryptite ss. The proto β -phase was formed over a narrow temperature range (600-700°C) and rapidly transformed into its β'_{II} -modification which showed a wider stability range upto 1040°C and then transformed into the stable δ_0 - $\text{Li}_2\text{ZnSiO}_4$ modification. The metastable β -eucryptite ss was

first formed instead of β -spodumene and starts its transformation into β -spodumene around 800°C .

γ_0 -lithium magnesium orthosilicate, β -eucryptite ss, β -spodumene, lithium metasilicate and forsterite were encountered during crystallization of $\text{Li}_2\text{MgSiO}_4$ - $\text{LiAlSi}_2\text{O}_6$ glasses.. The γ_0 - $\text{Li}_2\text{MgSiO}_4$ phase showed a limited and narrow stability range and dissociated rapidly around 800°C . The β -eucryptite ss \longrightarrow β -spodumene transformation was delayed with increasing the lithium magnesium component in the glass.

TiO_2 exhibits a catalytic effect on the formation of the proto β - $\text{Li}_2\text{ZnSiO}_4$ phase, the transformation of β -eucryptite ss to β -spodumene and the decomposition of the γ_0 - $\text{Li}_2\text{MgSiO}_4$. ZrO_2 showed a retarding effect on the formation of these phases.

The type, stability and compatibility relations of the crystallized phases are discussed in relation to glass composition and thermal treatment applied. Glass-ceramic with favourable properties could be obtained by controlled crystallization of glasses in these investigated $\text{Li}_2\text{ZnSiO}_4$ - $\text{LiAlSi}_2\text{O}_6$ and $\text{Li}_2\text{MgSiO}_4$ - $\text{LiAlSi}_2\text{O}_6$ systems.

-0000000000-

*

PART ONE
INTRODUCTION AND EXPERIMENTAL

CHAPTER 1

SCIENTIFIC BACKGROUND AND LITERATURE REVIEW

1.1. INTRODUCTION :

When glasses are cooled from the liquid state, they pass through a temperature range wherein devitrification may occur. This uncontrolled crystallization (devitrification), which arises at interfacial glass boundaries or from a small number of nucleation sites in the interior, may result in coarse-grained microstructures that may be accompanied by planes of weakness and gross distortion leading to low strength resultant materials. Glass formation implies the avoidance of such uncontrolled crystallization during cooling of the melt.

On the other hand, if the crystallization process could be controlled by adopting special glass formulations, suitable nucleation catalysts and proper heat treatment schedules, a new class of engineering materials (glass-ceramics) with desirable chemical, physical and mechanical properties could be obtained. Glass-ceramics are characterized by uniform fine-grained microstructures with randomly oriented crystals and some residual interstitial glassy matrix with

no voids or microcracks. Such structural characteristics provide improved performance and reliability of the glass-ceramics over traditional ceramics⁽¹⁻⁴⁾. As a result of this unique microstructure, glass-ceramics with properties such as translucency, high strength and very low and uniform thermal expansion can be routinely produced. Thus controlled crystallization has a number of key advantages.

Glass-ceramics, therefore, are primarily silicate-based materials formed by the usual glass-forming processes and converted through controlled crystallization into polycrystalline ceramic-like products by the proper heat-treatment. These glass-ceramics, which are also referred to in the technical literature as pyroceram, vitroceram, devitroceram, sitall, and melt-cast ceramics, generally have a crystal content greater than 40-50%.

The choice of glass composition, raw materials, and processing methods of glass manufacture and the final properties are to some degree variable so as to suit particular applications. The choice of composition for a glass-ceramic must depend not only on the ease with which the glass can be prepared and nucleated, but also on the properties expected

for the subsequent crystalline phases developed (3). A type of these phases, with a very low thermal expansion, is that based on the β -quartz structure. A variety of names has been proposed to identify the family of phases based on the β -quartz structure. The name " β -eucryptite ss"* is one of them used to indicate that Li^+ ions are the main stuffing ions present. Another type of the crystallization products which may yield useful glass-ceramics are the orthosilicate $\text{Li}_2\text{O} \cdot (\text{ZnO}, \text{MgO}) \cdot \text{SiO}_2$ phases. These orthosilicate phases are sensitive to the phase composition and can crystallize readily from a wide range of compositions. Complexities usually arise from their extensive polymorphism and wide ranges of solid solutions. For instance, $\text{Li}_2\text{ZnSiO}_4$ exhibits ten polymorphs. Phase transformations could be affected in reasonable times at low temperatures, although the transformations were often metastable.

Moreover, different varieties of metastable and stable polymorphs such as β -eucryptite ss, β -spodumene could also be encountered in the complex $\text{Li}_2\text{O} \cdot \text{Al}_2\text{O}_3 \cdot (\text{ZnO}, \text{MgO}) \cdot \text{SiO}_2$ system.

* ss = Solid solution(s).

On the other hand, as the number of possible phases in a system increases, the probability of production of fine-grained textures increases as well, because the complex nature of the chemical composition hinders the diffusion of ions over large distances. Therefore, the formation of many fine crystals is more favourable than the formation of few coarse crystals of the same phase. In addition, compositions crystallizing to give solid solutions are of importance with regard to the physical properties of the resultant materials. Crystallization of a glass to yield one or more solid solution series leads to a reduction in the grain-boundary energies and elimination of the stresses found at these boundaries consequently an increase in the strength of the material would result⁽⁵⁾. However, the more complex the composition of the glass is, the greater will be the diversity of the expected stable and metastable phases and the higher is the probability of a successful combination of properties in the resultant glass-ceramic⁽⁶⁾.

In order to increase the relevance of our studies to the chemically more complex commercial glass ceramics, a number of compositions within the systems spodumene ($\text{LiAlSi}_2\text{O}_6$)-lithium zinc orthosilicate ($\text{Li}_2\text{ZnSiO}_4$) and spodumene-lithium magnesium orthosilicate ($\text{Li}_2\text{MgSiO}_4$), were selected for study.

1.2. SCIENTIFIC BACKGROUND :

1.2.1. Crystallization of Glass :

Crystallization is the process by which the regular lattice of the crystal is generated from the less ordered liquid or glassy states⁽³⁾. It is generally considered as consisting of more or less two independent processes:

i) Nucleation, or formation of crystal centres, and ii) crystal growth on the formed centres⁽⁷⁾.

i) Nucleation :

For crystallization to begin, crystal nuclei must be present. Nucleation involves the initiation of regions of longer range atomic order, known as embryos, than are normally present in the molten state or in the supercooled liquids⁽⁸⁾. When these embryos attain a critical minimum size capable of developing spontaneously into gross particles of the stable phase they are known as nuclei.

Nucleation may take place either homogeneously, i.e. freely in the volume of the original phase, or heterogeneously on the surfaces of the container, surfaces of

foreign particles or on structural imperfections⁽⁹⁾. In homogeneous nucleation, the composition of the primary nuclei does not differ from that of the main crystalline phase, whereas in heterogeneous nucleation the crystallization of the glass is induced by introducing foreign nuclei. The nucleant, which is generally a metal, oxide or fluoride, is incorporated in the batch and becomes an integral part of the glass during melting.

However, despite considerable research, the mechanisms by which nucleating agents induce crystallization are not yet completely clear. A generalized picture of the nucleating mechanism cannot be developed, since the role of nucleating agents in catalyzing the formation of nuclei and the major crystalline phases undoubtedly differs from one nucleant to another. However, it is more safe to say that, by a mean or another, the nucleant causes a disproportionation of the binding forces⁽¹⁰⁾ which, in turn, introduce sites of lower thermodynamic stability.

Stookey⁽¹⁾ listed some of the requirements of an effective nucleating agent, viz. a) it should be readily